

[o.m13.p9] Phase analysis of Na_xWO_3 and $\text{Na}_x\text{Mo}_y\text{W}_{1-y}\text{O}_3$.

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In recent years many research groups have studied tungsten bronzes, M_xWO_3 and mixed bronzes, $\text{M}_x\text{M}'_y\text{W}_{1-y}\text{O}_3$ ⁽¹⁻⁴⁾, where $\text{M}' = \text{V}, \text{Nb}, \text{Mo}$ or Ta because of their remarkable physical and chemical properties ⁽⁵⁾.

Sodium tungsten bronzes, $\text{Na}_{0.60}\text{WO}_3$ and molybdenum substituted sodium tungsten bronzes, $\text{Na}_{0.60}\text{Mo}_{1-y}\text{WO}_3$, where $y=0.05-0.20$, were prepared from the appropriate amount of different oxides at $600^\circ\text{C}-800^\circ\text{C}$ for a week in evacuated sealed silica tubes. The samples were characterized by X-ray powder technique using Guinier-Hägg focusing camera.

The prepared samples were indexed as perovskite tungsten bronze, PTB (cubic) type. It was observed that the tungsten atoms could be substituted up to 15% by molybdenum atoms in sodium PTB (cubic) phases at 600°C and 700°C .

A refinement of cell parameter in molybdenum substituted sodium PTB (cubic) phase showed an increase in cell parameter with increasing molybdenum content in $\text{Na}_{0.60}\text{Mo}_{1-y}\text{WO}_3$.

[o.m13.p10] Structure and low temperature yield stress of quenched Al-Li alloys.

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The theory of thermally activated overcoming of local impurity barriers by dislocations predicts a monotonic increase in effective stress τ^* with decreasing temperature T and increasing impurity concentration C in an equilibrium solid solution. To test this hypothesis, temperature dependences of the yield stress, other parameters of plasticity and structure of the Al-Li quenched alloys were studied at temperatures ranged from 4.2 to 290 K. The experimental data show that at $T < 140$ K the value of τ^* does increase with C for a Li content of $C < 7$ at.% but decreases on further doping. The relation between the lattice parameter and initial Li concentration in the alloy studied by the X-ray diffraction technique with averaging the data by nine reflections demonstrates that the true Li concentration in the alloy matrix is lower than the initial one. Moreover, the total X-Ray diffraction patterns of the concentrated alloy display new diffraction peaks. The X-ray diffraction data suggest that the concentrated alloy is not a single-phase one: even with quenching, there occur dispersed precipitations of the second phase. Thus, the decrease in τ^* in the concentrated alloy which is contradictory to the theory can be accounted for the effect of second phase precipitations on kinetics of thermally activated motion of dislocations in the matrix.

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