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MS19-O5

Investigating CO₂ uptake in Sc₂BDC₃ using XRD, *ab initio* DFT and GCMC methods

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Despite the large number of MOFs that have been synthesised and are present in the Cambridge Structural Database, 84,185 as of Nov. 2017, very few have been studied for both their gas adsorption and mechanical properties.¹ Most investigations concerning the former are based upon adsorption isotherms and hence there is a lack of understanding concerning the location of adsorbed guest molecules within the pores and the nature of specific interactions between the guest molecules and the framework. Up to 2014, 120 crystallographic studies on frameworks containing adsorbed gas molecules had been reported, with only 47 frameworks investigated in all.² A more detailed understanding of these interactions will help us develop more insightful structure-property relationships, thereby aiding chemists in designing MOFs with enhanced or specific guest uptake.

The small-pore MOF Sc₂BDC₃ (where BDC = benzenedicarboxylate) was initially observed to undergo an orthorhombic-to-monoclinic phase transition, via rotation of one pair of BDC linkers, under CO₂ uptake at $p = 1$ bar and 235 K.³ Here, we have used a combination of capillary gas cell XRD experimental and *ab initio* DFT and GCMC computational methods to show that the framework also undergoes the same phase transition, gradually, at 298 K at higher CO₂ pressures (complete at 3 bar). Additionally, a third adsorption site, which was not seen in the original investigation at 235 K, was observed; this discovery verified the expected maximum uptake of CO₂.

Ab initio DFT calculations determined that the monoclinic form of Sc₂BDC₃ is the lower energy geometry-optimised structure (by 13.3 kJ mol⁻¹); the orthorhombic phase however is present at 298 K due to the entropic term having greater weighting at this temperature. Therefore, the formation of the monoclinic phase at room temperature upon CO₂ uptake is a result of CO₂-framework interactions exceeding the energy barrier to linker rotation. This was shown using GCMC to model CO₂ uptake in the experimental framework structures, with stronger CO₂-framework interactions in the monoclinic phase. Additionally, there was good agreement between the binding site hierarchies observed by the three different methods, with a clear switching of the site hierarchy between the two phases. When a CO₂/CH₄ gas mixture was applied to this material, the XRD and computational methods showed that Sc₂BDC₃ has selectivity for CO₂ over CH₄; this was a result of weaker CH₄-framework interactions and specific CH₄ adsorption site locations which allowed CO₂ to occupy its strongest adsorption site unhindered.

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MS20 Rating and improving data quality: instrumentation, analysis and postprocessing

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MS20-O1

Experimental charge density studies: improvements in data processing

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In experimental charge density investigation, it is indispensable to use the highest possible quality of data. Systematic errors should be avoided.

One systematic error is the low-energy contamination caused by focusing multilayer optics. To eliminate this problem filtering by interposition of a low-density material foil into the beam was proposed. However, filtering lowers the intensity, which might be crucial, especially for charge density data collection, and is of course not possible for already measured data. The low-energy contamination mainly affects the reflections with indices 3h 3k 3l. In the program SADABS [1], a 3 λ correction in analogy to the earlier 2/ λ correction is now implemented. [2]

In several charge density studies, we noticed that the residual density improved significantly after ten resolution dependent scale factors were refined. The dependency of the scale factors against resolution showed a typical pattern for all data sets, but the variation was much smaller for data collected at 15 K instead of 100 K. Therefore, this procedure seems to cure errors that are resolution and temperature dependent such as thermal diffuse scattering (TDS). Similar results could be obtained by processing data with relatively small integration boxes instead of refined box sizes. This procedure seems to emulate a rough TDS correction. However, to find the best integration box is very time consuming. In a nested interval approach, a correction factor $\{\alpha = a[\sin(\vartheta)/\lambda]^2 + b[\sin(\vartheta)/\lambda]^3\}$ is determined that minimizes these errors and improves the model quality. [3]

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